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**Spin orders in the charge disproportionated phases of A-site layer ordered triple perovskite  $\text{LaCa}_2\text{Fe}_3\text{O}_9$ .**

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**Abstract.**

The coupling between spins and charge disproportionation states has been investigated in the  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  oxide with neutron powder diffraction. This A-site layer ordered triple perovskite  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  undergoes charge disproportionation on cooling and shows two different charge ordering patterns. At 230 K,  $\text{Fe}^{3.67+}$  disproportionates into a 2:1 ratio of  $\text{Fe}^{3+}:\text{Fe}^{5+}$  which order in a layered manner along the  $\langle 010 \rangle$  direction of the pseudocubic unit cell. At lower temperatures ( $T < 170$  K), the charge ordering pattern changes to a layered arrangement along the  $\langle 111 \rangle$  direction. Neutron powder diffraction data shows that in the intermediate temperature range ( $170 \text{ K} < T < 230 \text{ K}$ ) the spins order into a cycloidal structure on the  $ac$  plane for the  $\text{Fe}^{3+}$  cations while the  $\text{Fe}^{5+}$  cations remain paramagnetic. For the

lowest temperature range ( $2 \text{ K} < T < 190 \text{ K}$ ), the spin structure follows the charge ordering and evolves to a  $\langle 111 \rangle$  layered magnetic structure.

## **Introduction.**

Charge ordering is often observed in transition-metal-oxide systems with electronic instabilities of mixed valent cations. The Verwey transition of  $\text{Fe}_3\text{O}_4$  with  $\text{Fe}^{2+}/\text{Fe}^{3+}$  order is a famous example and the phenomenon is evidenced by a drastic change in transport properties.<sup>1,2,3</sup> Similar charge ordering behaviours are observed in a few oxides containing unusually high valent cations due to charge disproportionation (CD) of  $\text{Fe}^{4+}$ . The perovskite  $\text{CaFeO}_3$  shows a metal-insulator transition at 290 K due to CD of  $\text{Fe}^{4+}$  to  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  which order in a rock-salt type arrangement. The CD transition in  $\text{CaFeO}_3$  changes the uniform  $\text{Fe}^{4+}\text{O}_6$  octahedra to alternating large  $\text{Fe}^{3+}\text{O}_6$  and small  $\text{Fe}^{5+}\text{O}_6$  octahedra, thus the charge order is coupled with the lattice degree of freedom.<sup>4,5</sup> Other examples of CD in Fe-containing perovskite-structure oxides include rock-salt ordered  $\text{Fe}^{3+}/\text{Fe}^{5+}$  in the A-site ordered  $\text{CaCu}_3\text{Fe}_4\text{O}_{12}$  and checkerboard ordered  $\text{Fe}^{3+}/\text{Fe}^{5+}$  in the B-site layer ordered  $\text{Ca}_2\text{FeMnO}_6$ .<sup>6,7</sup> The CD in high-valent Fe has been rationalized as the localization of ligand holes which are produced by strong hybridization of Fe  $d$  with O  $p$  orbitals.<sup>8,9</sup>

An important aspect of CD behavior concerns coupling of the charge and spin degrees of freedom. In  $\text{CaFeO}_3$  a magnetic transition occurs at 210 K, below the CD transition temperature, and a helical magnetic structure of  $\text{Fe}^{3+}/\text{Fe}^{5+}$  spins is stabilized.<sup>5</sup> In the A-site ordered quadruple perovskite  $\text{CaCu}_3\text{Fe}_4\text{O}_{12}$ , on the other hand, the magnetic transition temperature is the same as the CD transition temperature, and the  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  spins couple ferromagnetically below the CD transition temperature giving rise to ferrimagnetism when coupled with the Cu ions.<sup>10,11</sup> In the B-site layered double perovskite  $\text{Ca}_2\text{FeMnO}_6$ , three-

dimensional noncollinear magnetic order develops below 95 K, although the two-dimensional CD of  $\text{Fe}^{4+}$  occurs at 200 K.<sup>7</sup> As can be observed, the coupling between the charge and spins in the CD transition varies significantly between oxides containing high-valent Fe ions.

Very recently we found that a new A-site layered triple perovskite  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  shows CD from high-valent  $\text{Fe}^{3.67+}$  to a 2:1 ratio of  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$ .<sup>12</sup> Interestingly, the compound shows two successive structural transitions due to different CD ordering patterns. The first transition occurs at 230 K, where the charge disproportionated 2:1  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  are ordered in a layered manner, along the  $\langle 010 \rangle$  layer stacking direction of the pseudocubic unit cell. Below the second transition at 170 K, the charge disproportionated 2:1  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  are ordered along the  $\langle 111 \rangle$  direction of the pseudocubic unit cell. Magnetic susceptibility data collected as a function of temperature shows that both transitions occur at the same temperatures as the magnetic anomalies.<sup>12</sup> The simple perovskite  $\text{La}_{1/3}\text{Ca}_{2/3}\text{FeO}_3$  with identical chemical composition to that of the present  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  but with random arrangement of the A-site La and Ca ions presents only one structural transition at 217 K to a 2:1  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  charge ordering pattern along the  $\langle 111 \rangle$  direction of the pseudocubic perovskite unit cell, similar to the one found at the lowest temperatures in the A-site layer ordered analogue.<sup>13</sup> The magnetic structures of the CD states have not been determined.

In this study, we analyzed the magnetic structures of this newly discovered layered compound which shows unusual CD behaviors. Using neutron powder diffraction data, we have revealed that the two charge ordering patterns result in different magnetic structures. The magnetic structures of  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  and  $\text{La}_{1/3}\text{Ca}_{2/3}\text{FeO}_3$  are compared to observe the effects of the cation ordering upon the spin orders.

## Experimental.

The synthesis of  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  and  $\text{La}_{1/3}\text{Ca}_{2/3}\text{FeO}_3$  was carried out as reported previously.<sup>12,13</sup> Powder neutron diffraction patterns from both samples were measured using the time-of-flight neutron diffractometer WISH at the ISIS spallation source.<sup>14</sup> Data were collected every 50 K between 2 and 300 K and every 5 K in the  $150 \text{ K} < T < 250 \text{ K}$  temperature range on warming. All the data were normalised using the MantidPlot program.<sup>15</sup> Crystal and magnetic structures were refined using Fullprof suite software package.<sup>16</sup> Symmetry analysis was performed with BasIreps within the same software.

## Results and discussion.

Figure 1a shows the neutron diffraction patterns (Bank 2) as a function of temperature from 250 to 150 K collected using the WISH diffractometer on warming. The evolution of two different phases due to the successive transitions at 230 and 170 K reported in our previous study can clearly be seen and the patterns are coloured in red and blue respectively.<sup>12</sup>

Neutron diffraction data collected at 300 K was fitted well using the structure reported previously.<sup>12</sup> It consists of a  $a \square = a + b$ ,  $b \square = 6c$ ,  $c \square = -a + b$   $Pnma$  supercell model (with  $|a| = |b| = |c| = a_p$  being the lattice parameter of the simple perovskite structure  $\approx 3.8 \text{ \AA}$ ), necessary to account for the 2:1 layered A-site ordering between Ca and La and the octahedral tilting. The refined occupancies for the La and Ca are 0.67(1)/0.33 and 0.18(1)/0.82 for the different A sites, in agreement with our SXRD results, thus confirming the A-site layered structure. The structure is reported in Supplementary Table 1. The absence of vacancies at any of the oxygen sites is also confirmed by the neutron refinement.

The structural model accounts for all the observed reflections in data collected from  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  at temperatures between  $240 \text{ K} \leq T \leq 300 \text{ K}$ . Figure 2 top shows the long  $d$ -

spacing bank of the diffraction data at 240 K; other banks and R-factors are shown in Supplementary Information (Figure S1). Although additional superstructure diffraction peaks appear below 230 K, the nuclear diffraction peaks originating from the *Pnma* supercell fit well the essential diffraction maxima and show a cell-volume decrease at 230 K and an increase at 190 K (Figure 1b). These temperature dependent changes in the crystal structure are consistent with those observed in SXRD data in our previous study.<sup>12</sup>

At least five additional diffraction maxima were clearly observed when the sample was cooled below 230 K as shown in Figure 1a. These features are not present in SXRD data collected from  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  in the temperature range  $170 \text{ K} < T < 230 \text{ K}$ , suggesting the origin of these peaks is likely magnetic.<sup>12</sup> Mössbauer spectroscopy data collected in this temperature range show the presence of  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  in a 2:1 ratio and that the  $\text{Fe}^{3+}$  spins are magnetically ordered while the  $\text{Fe}^{5+}$  spins remain paramagnetic. Therefore, an additional magnetic model was included in the fit to the observed NPD in between  $190 \text{ K} < T < 230 \text{ K}$ .

All the additional magnetic reflections can be indexed with propagation vector  $k_1 = [1/3 \ 1/3 \ 1/3]$  on a cubic  $a_p \times a_p \times a_p$  cell, or with  $k_{1\Box} = [1/3 \ 0 \ 0]$  on the *Pnma* supercell that allows for the layered ordering of  $\text{La}^{3+}$  and  $\text{Ca}^{2+}$  cations. As mentioned earlier, our Mössbauer study shows that the  $\text{Fe}^{5+}$  spins are not long range ordered (idle) in this temperature range; thus, the  $\text{Fe}^{3+}$  spins are ordered below the CD transition temperature, whereas the  $\text{Fe}^{5+}$  spins remain paramagnetic. We performed symmetry analysis, [17] Irreducible Representations are reported in Supplementary Table 2. The best refined magnetic structure ( $R_{\text{mag}} = 2.9 \%$  for Bank 2) consists of a spin cycloid with  $120^\circ$  rotation along the (100) direction and a magnitude of  $1.96(2) \mu_B$  per  $\text{Fe}^{3+}$  as shown on Figure 3a. The  $\text{Fe}^{3+}$  magnetic layers are AFM coupled along the *b* axis and the spins are confined to the *ac* plane. Spin density wave-like solutions are also plausible with similar R factors as the cycloid model and cannot be distinguished with powder data, however there are no obvious reasons

for the moments to take different values and Mössbauer results support only one magnetic moment value for the  $\text{Fe}^{3+}$ , therefore the spin cycloid was preferred.  $\text{Fe}^{5+}$  spins were constrained to zero values since refined moments were in the limit of detection ( $\approx 0.4 \mu_{\text{B}}$ ) and no significant improvement of the R factors was observed, this is also in accordance to Mössbauer data. After refinement of the magnetic structure, it was analysed with FINDSYM [18], the resulting magnetic space group is  $Pmc2_1$ . It is interesting that both the charge ordering pattern and the magnetic structure show two-dimensional features reflecting the A-site layered crystal structure.

Neutron powder diffraction data collected below  $T_{\text{N2}} = 190 \text{ K}$ , where the second structural transition due to the charge ordering pattern change from the  $\langle 010 \rangle$  to  $\langle 111 \rangle$  direction occurs, show additional reflections which increase in intensity down to the lowest temperature measured (1.5 K). These peaks were also assumed to be magnetic since our previous Mössbauer results indicated that both  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  spins are magnetically ordered at these temperatures and no additional reflections were observed in SXR data. These reflections can be indexed with a propagation vector  $k_2 = [1/6 \ 1/6 \ 1/6]$  on the  $a_{\text{p}} \times a_{\text{p}} \times a_{\text{p}}$  cubic cell. To account for the charge ordering pattern, the refinement was performed with an equivalent description in a  $a_{\text{p}} \times a_{\text{p}} \times c_{\text{p}}$  hexagonal cell with  $a_{\text{p}} = a - b$ ,  $b_{\text{p}} = b - c$  and  $c_{\text{p}} = a + b + c$ ,  $\alpha = \beta = 90^\circ$ ,  $\gamma = 120^\circ$  hexagonal cell with a propagation vector  $k_{2\text{p}} = [0 \ 0 \ 1/2]$ . Symmetry analysis was performed (Supplementary Table 4) and the best refined low temperature magnetic structure model is shown in Fig. 3b ( $R_{\text{mag}} = 5.9 \%$  for Bank 2). The spins of the charge disproportionated  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  ions, which adopt a 2:1 layered arrangement along the  $\langle 111 \rangle$  direction of the pseudocubic unit cell, are also ordered in a collinear way. The refined moments at 1.5 K are 3.0(1) and 2.4(1)  $\mu_{\text{B}}$  for  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  respectively. The spins of  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  within each layer are ferromagnetically (FM) ordered and this triple-layered spin structure is AFM stacked along the same  $\langle 111 \rangle$  direction. Because the spin direction of every

triple layer is opposite, no net magnetization is given with this magnetic structure, which is consistent with the observed magnetic susceptibility at low temperatures.<sup>12</sup> The magnetic space group resulted in  $C2/m$  within a  $-a \times -2b \times -a, 2c$  monoclinic cell. Temperature dependencies of the magnetic moments are shown on Figure 3c. An important point is that the  $\text{Fe}^{3+}/\text{Fe}^{5+}$  magnetic structure drastically changes when the CD pattern changes.

As we reported previously, the A-site disordered analogue  $\text{La}_{1/3}\text{Ca}_{2/3}\text{FeO}_3$  shows the same charge ordering pattern along the  $\langle 111 \rangle$  direction below 217 K.<sup>13</sup> As shown in Fig. S2b, neutron powder diffraction collected at 1.5 K of this disordered analogue is similar to that observed from  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  and the data are well accounted for using an identical magnetic structure ( $R_{\text{mag}} = 4.04\%$  for Bank 2). The refined moments at 1.5 K are 3.4(1) and 2.4(1)  $\mu_{\text{B}}$  for  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  respectively. Therefore, the ground state CD patterns and magnetic structures are the same for the A-site layer ordered  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  and the A-site disordered  $\text{La}_{1/3}\text{Ca}_{2/3}\text{FeO}_3$ . The results also imply that the ground state magnetic structures are primarily determined by the arrangement of  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  ions and their magnetic interactions, and are less influenced by the A-site order/disorder.

The unusual magnetic structure observed at intermediate temperatures, where only  $\text{Fe}^{3+}$  spins are ordered while the  $\text{Fe}^{5+}$  spins remain paramagnetic, is only stabilized in the A-site layer ordered compound. Taking into account the charge ordering pattern at the intermediate temperatures, which implies a 2:1 order of  $\text{Fe}^{3+}:\text{Fe}^{5+}$  along the  $\langle 010 \rangle$  direction; a simple analysis with Goodenough-Kanamori-Anderson rules (GKA) explains the unusual spin cycloid magnetic structure. Considering the layered charge ordering pattern, the strongest interaction along the  $b$  axis is AFM between  $\text{Fe}^{3+} - \text{Fe}^{3+}$  (half-filled  $e_g$  orbitals) and frustrates the FM coupling to the  $\text{Fe}^{5+}$  (empty  $e_g$  orbitals) due to the magnetic periodicity along this direction. A similar magnetic frustration has been recently observed in the

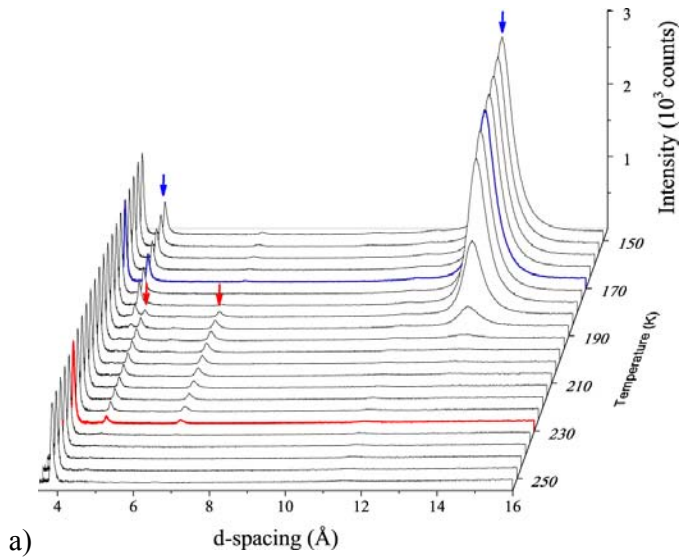


$\text{Ca}_{0.5}\text{Bi}_{0.5}\text{Fe}^{3.5+}\text{O}_3$  compound. This material shows CD of  $\text{Fe}^{3.5+}$  to a 2:1 ratio of  $\text{Fe}^{3+} : \text{Fe}^{4.5+}$  and the magnetic order shows AFM coupling between  $\text{Fe}^{3+}$  spins and idle  $\text{Fe}^{4.5+}$  spins. <sup>19</sup>

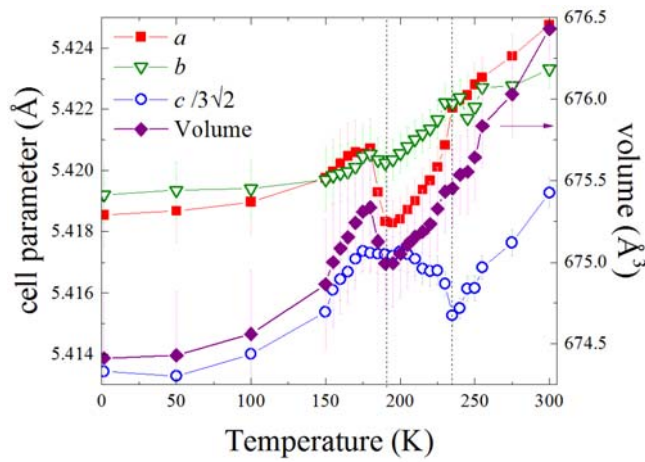
In conclusion, we have determined the intermediate and ground magnetic states in  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  which are both coupled to their respective CD pattern. The intermediate magnetic state shows an unusual spin cycloid with AFM layers of  $120^\circ$  rotating spins of  $\text{Fe}^{3+}$  and idle  $\text{Fe}^{5+}$  spins. The ground state on the contrary shows triple FM layers stacked AFM along the  $\langle 111 \rangle$  direction. These magnetic structures are explained with the magnetic interactions according to the GKA rule. The disordered  $\text{La}_{1/3}\text{Ca}_{2/3}\text{Fe}_3\text{O}_9$  presents the same magnetic ground state. The A-site order is responsible for the intermediate phase since this is not observed in the disordered analogue; however, the ground state is very robust as the same magnetic structure is observed in both ordered and disordered compounds.

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a)



b)

Figure 1. a) Thermal evolution of the NPD patterns between 150 and 250 K on warming. Magnetic Bragg reflections are marked with arrows, the coloured patterns show the transition temperatures observed by magnetometry at 230 K and 170 K, red and blue arrows respectively. b) Temperature dependence of the lattice parameters and the volume from NPD data of  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  with both CD/magnetic transitions marked by dashed lines. The transition temperatures appear at 190 K and 235 K here. They differ from the previously reported 170 K and 230 K values due to possible differences in warming rate or thermal equilibration in the previous experiments.

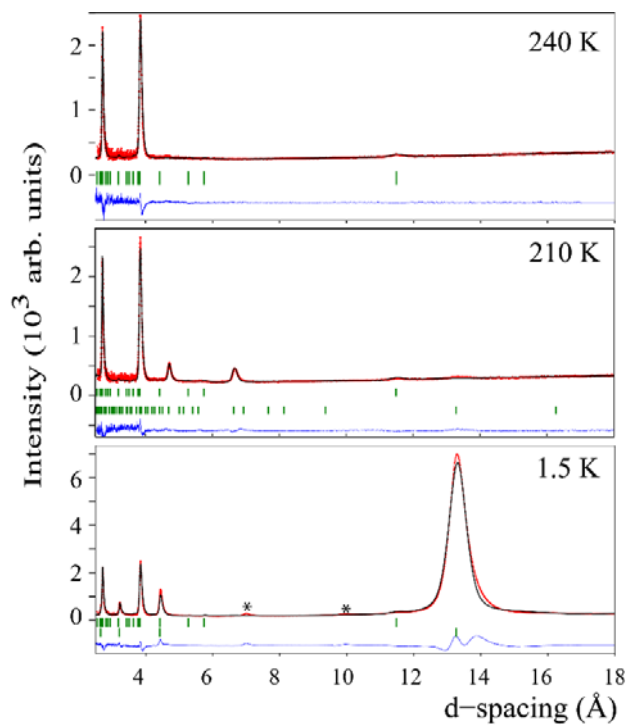


Figure 2.  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  Rietveld refinement plots of neutron powder-diffraction data collected on WISH@ISIS at 240 K, 210 K and 1.5 K. The observed (red dots), calculated (black lines) and difference (blue lines) patterns are shown. The allowed Bragg reflections (green vertical lines) for nuclear (top) and magnetic (bottom) phases are shown. Small asterisks mark magnetic diffraction from a secondary  $\text{CaFeO}_3$  phase.

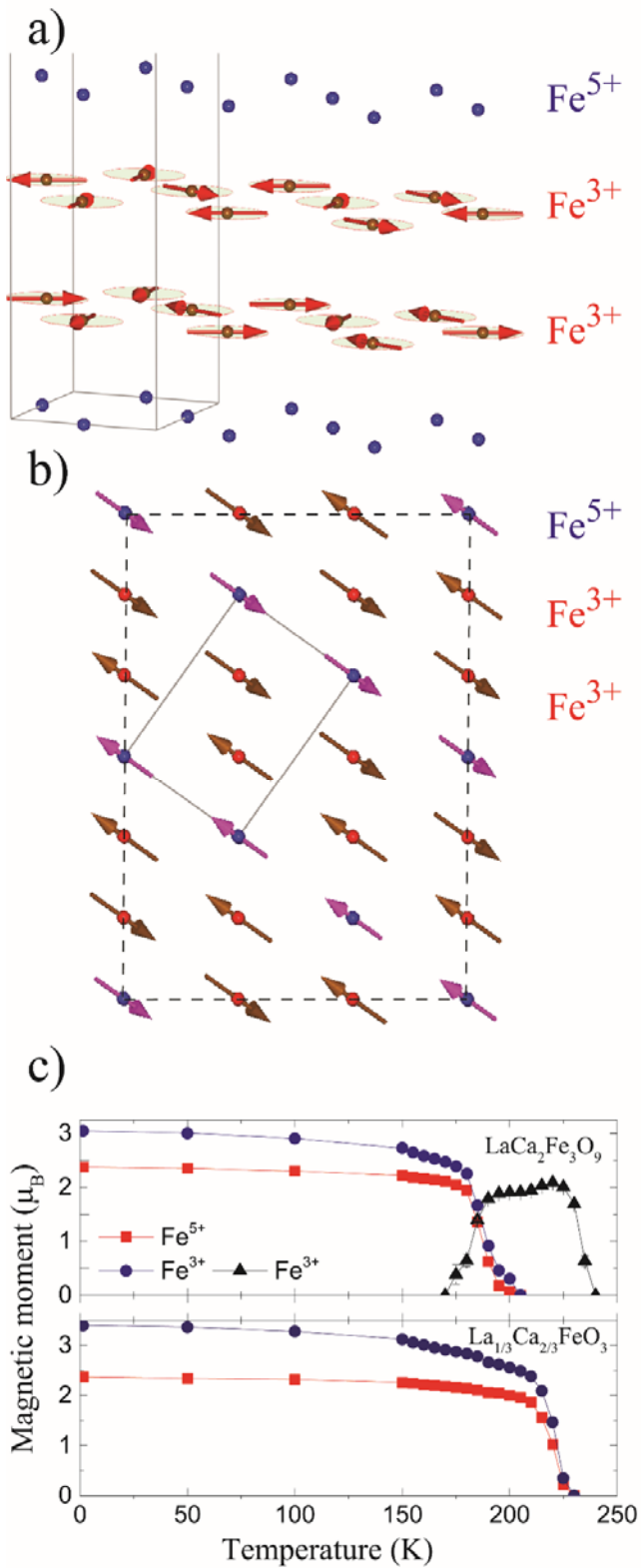


Figure 3. a) Magnetic structure of  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  in the intermediate phase,  $170 \text{ K} < T < 240 \text{ K}$ .  $\text{Fe}^{5+}$  spins remain idle while  $\text{Fe}^{3+}$  spins order with a propagation vector  $k_{I\Box} = (1/3, 0, 0)$  on a

*Pnma* cell (only half of the cell is shown). b) Low temperature magnetic structure of  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  ( $1.5 \text{ K} < T < 200 \text{ K}$ ) and magnetic structure of  $\text{La}_{1/3}\text{Ca}_{2/3}\text{FeO}_3$  ( $T_N = 217 \text{ K}$ ). Both  $\text{Fe}^{3+}$  and  $\text{Fe}^{5+}$  order on a triple layered antiferromagnetic structure with a propagation vector  $k_{2\Box} = (0, 0, \frac{1}{2})$  on a  $\sqrt{2}a_p \times \sqrt{2}a_p \times \sqrt{3}a_p$  hexagonal cell shown with lines. Dashed lines show the  $3\sqrt{2}a_p \times 3\sqrt{2}a_p \times 6a_p$  supercell needed to account for the CD 2:1  $\text{Fe}^{3+}:\text{Fe}^{5+}$ , layered A-site cation ordering and octahedral tilting. c) Magnetic moment as function of temperature obtained from Rietveld refinement on  $\text{LaCa}_2\text{Fe}_3\text{O}_9$  and the disordered  $\text{La}_{1/3}\text{Ca}_{2/3}\text{FeO}_3$ .

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<sup>1</sup> E. J. W. Verwey, *Nature* **144**, 327–328 (1939).

<sup>2</sup> E. J. W. Verwey and P. W. Haayman, *Physica* **8**, 979 (1941).

<sup>3</sup> M. S. Senn, J. P. Wright and J. P. Attfield. *Nature* **481**, 173 (2012).

<sup>4</sup> M. Takano, N. Nkanishi, Y. Takeda, S. Naka and T. Takada. *Mater. Res. Bull.* **12**, 923 (1977).

<sup>5</sup> P. M. Woodward, D.E. Cox, E. Moshopoulou, A. W. Sleight and S. Morimoto. *Phys. Rev. B.* **62**, 844 (2000).

<sup>6</sup> I. Yamada, K. Takata, N. Hayashi, S. Shinohara, M. Azuma, S. Mori, S. Muranaka, Y. Shimakawa and M. Takano. *Angew. Chem. Int. Ed.* **47**, 7032 (2008).

<sup>7</sup> Y. Hosaka, N. Ichikawa, T. Saito, P. Manuel, D. Khalyavin, J.P. Attfield and Y. Shimakawa. *J. Am. Chem. Soc.* **137**, 7468 (2015).

<sup>8</sup> W-T. Chen, T. Saito, N. Hayashi, M. Takano and Y. Shimakawa. *Sci. Rep.* **2**, 449 (2012).

<sup>9</sup> Y. Shimakawa, *J. Phys. D. Appl. Phys.* **48**, 504006 (2015).

<sup>10</sup> M. Mizumaki, W. T. Chen, T. Saito, I. Yamada, J. P. Attfield, and Y. Shimakawa, *Phys. Rev. B* **84**, 094418 (2011).

<sup>11</sup> Y. Shimakawa and M. Mizumaki, *J. Phys. Cond. Matter.* **26**, 473203 (2014).

<sup>12</sup> H. Guo, Y. Hosaka, F. D. Romero, T. Saito, N. Ichikawa and Y. Shimakawa. *Inorg. Chem.* **56**, 3695 (2017).

<sup>13</sup> H. Guo, Y. Hosaka, H. Seiki, T. Saito, N. Ichikawa and Y. Shimakawa. *J. Solid State Chem.* **246**, 199 (2017).

<sup>14</sup> L. C. Chapon, P. Manuel, P. G. Radaelli, C. Benson, L. Perrot, S. Ansell, N. J. Rhodes, D. Raspino, D. Duxbury, E. Spill and J. Norris. *Neutron News* **22**, 22 (2011).

<sup>15</sup> O. Arnold, J. C. Bilheux, J. M. Borreguero, A. Buts, S. I. Campbell, L. Chapon, M. Doucel, N. Draper, R. Ferraz Leal, M. A. Gigg, V. E. Lynch, A. Markvarsdén, D. J. Mikkelsen, R. Miller, K. Palmen, P. Parker, G. Passos, T. G. Perring, P. F. Peterson, S. Ren, M. A. Reuter, A. T. Savici, J. W. Taylor, R. J. Taylor, R. Tolchenov, W. Zhou and J. Zikovsky. *Nuclear Instruments and Methods in Physics Research A* **764**, 156 (2011).

<sup>16</sup> J. Rodriguez-Carvajal. *Physica B* **192**, 55 (1993).

<sup>17</sup> E. F. Bertaut. *Acta Cryst.* **A24**, 217 (1968).

<sup>18</sup> H. T. Stokes and D. M. Hatch, *J. Appl. Cryst.* **38**, 237 (2005).

<sup>19</sup> F. Denis Romero, Y. Hosaka, N. Ichikawa, T. Saito, G. McNally, J. P. Attfield, and Y. Shimakawa, *Phys. Rev. B* **96**, 064434 (2017).